Betavoltaics using scandium tritide and contact potential difference

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Tritium-powered betavoltaic micropower sources using contact potential difference (CPD) are demonstrated. Thermally stable scandium tritide thin films with a surface activity of 15 mCi/cm² were used as the beta particle source. The electrical field created by the work function difference between the ScT film and a platinum or copper electrode was used to separate the beta-generated electrical charge carriers. Open circuit voltages of 0.5 and 0.16 V and short circuit current densities of 2.7 and 5.3 nA/cm² were achieved for gaseous and solid dielectric media-based CPD cells, respectively. © 2008 American Institute of Physics. [DOI: 10.1063/1.2887879]

Recently, miniaturized radioisotope micropower sources have drawn significant attention because of their potential of providing a long-term energy solution for niche low-power consuming microsystems for a range of applications including exploration and surveillance.^{1–6} A number of nuclear-to-electrical energy conversion integrations, which had been investigated previously, are receiving renewed attention and novel materials and modern microfabrication techniques are being introduced.^{1,3} In recent years, a series of studies have been reported on betavoltaics employing semiconductor *pn* junction using materials such as amorphous silicon,² porous silicon,⁴ silicon carbide,^{5,6} and gallium arsenide/nitride.¹ The efficiency and longevity of these battery structures rely on high-quality *pn* junctions, which are often susceptible to radiation damage and lack structural self-healing resiliency.

One simple method of converting the kinetic energy of ionizing radioemissions into electrical energy is to utilize a contact potential difference (CPD) device produced from materials with dissimilar work functions. An appropriate dielectric medium, in the form of a thin film, gas, or liquid, is sandwiched between the two dissimilar electrodes. An ionizing radiation source, dispersed within the dielectric medium or included in an electrode, provides the energy source for electron-ion pair generation. Charge carriers are separated by the built-in electric field, created by the contact potential difference of the electrodes, generating current in an external load. A CPD-based micropower source is relatively simple to implement and can be integrated for on-chip applications.

Tritium, a radioisotope of hydrogen, is a good candidate for betavoltaic applications given its benign radiation characteristics, its relatively long half-life of 12.3 years, and availability.^{2,4,7,8} Tritium is a pure beta emitter producing energetic electrons with an average energy of 5.7 keV and a maximum energy of 18.6 keV. Considering that the threshold electron energy for disruption of the silicon lattice due to knock-on collisions is 20 keV,⁹ tritium decay beta particles pose little radiation damage concern for on-chip energy conversion devices. These factors make tritium an appealing power source for on-chip applications. Tritium in the gaseous form has been used in self-power lighting and in betavoltaic applications.^{1,4} However, the gaseous form of tritium has a low-power density of 87 μ W/cm³, not to mention the challenge of hermetic containment of tritium gas and associated hazard of tritium leakage. In this letter, we demonstrate a betavoltaic micropower source using high-density tritium metal hydride foils. Tritium is integrated in one of the electrodes in the form of a stable scandium tritide (ScT_x, where x is T to Sc atom ratio) thin film.

Fabrication of the ScT_x film commenced with the thermal evaporation of scandium; the Sc film, of 300 nm in thickness, is deposited on a silicon substrate. The Sc film was subsequently tritiated by exposing it to tritium gas for 10 h at a temperature of 250 °C and tritium pressure of 10 bars. The high-pressure tritium exposure apparatus used in this experiment is described elsewhere.⁸

To study the stability of the tritiatied ScT_x film, a thermal effusion measurement^{10,11} was performed. Effusion data showed tritium outgassing from the samples beginning at approximately 250 °C, the tritium loading temperature, while the majority of tritium evolution occurred near 600 °C. Thermal effusion experiments provided the concentration of tritium in the ScT_x films of approximately 60 mCi/cm², which is equivalent to a tritium to scandium atomic ratio x=1.

A 3.3 cm² ScT on Si sample was placed in the tritium effusion monitor, in ambient atmosphere at room temperature for a week to monitor the long-term stability of tritium in the film. An average outgassing rate of less than 2 pCi cm⁻² s⁻¹ or 4.2×10^7 atoms cm⁻² s⁻¹ was obtained. Assuming that the outgassing rate is undiminished with time, this is equivalent to a tritium diffusion half-life of greater than 100 years, over eight times the tritium decay half-life. The loss of tritium from the sample due to room temperature outgassing may be thus considered negligible.

Tritium immobilized in ScT film emits beta particles and the betas ionize the gas near the film. A beta particle, traversing in air, with an initial average kinetic energy of 5.7 keV produces approximately 170 electron-ion pairs. A surface activity monitor (SAM) was used to characterize the effective surface activity available for ionization.¹² A schematic of the

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FIG. 1. Ionization current density (J) due to a ScT film sample in air as a function of bias voltage (V). The air gaps are 1, 2, 4, and 6 mm, respectively.

SAM monitor is shown in the inset of Fig. 1; the ScT sample, of 2 cm² area, is placed on the lower electrode. The ionization current density (*J*) as a function of the applied bias voltage at electrode gaps of 1, 2, 4, and 6 mm is shown in Fig. 1. The saturated current density is approximately 15 nA/cm². Using the average ionization energy in air of 33.75 eV per ion pair, the effective surface activity of the ScT film is estimated to be 15 mCi/cm², which is equivalent to an average energy (5.7 keV) beta flux density of 90 pA/cm² or a power density of 0.5 μ W/cm².

To extract the energy of the beta particles, the contact potential difference of two dissimilar metal electrodes was employed to produce an electric field. The ScT film was used as the cathode, as well as the ionization source, while copper was used as the anode. The work functions of pure Sc and Cu are 3.5 and 4.7 eV, respectively. The active plate area was 2.5×1.3 cm² and the air gaps were varied from 0.05 to 3 mm. A Keithley 617 electrometer was used to measure the voltage and current. The measurement was carried out at room temperature and standard pressure. The IV characteristics of the CPD cell at electrode distances of 0.1, 0.3, and 0.6 mm are shown in Fig. 2. An open circuit voltage $(V_{\rm oc})$ of 0.5 V and a short circuit current $(I_{\rm sc})$ of 8.73 nA, which is equivalent to a short circuit current density of 2.67 nA/cm^2 , were measured at a distance of 0.1 mm. This current density is about 20% of the saturation current density at an air gap of 1 mm shown in Fig. 1, in recognition of the fact that the saturation current density drops significantly



FIG. 2. The *IV* characteristics of a 2.5×1.3 cm² CPD cell in air with electrode separations of 0.1, 0.3, and 0.6 mm, respectively.



FIG. 3. The short circuit current (I_{sc}) and the open circuit voltage (V_{oc}) as a function of the electrode separation (d) in a CPD cell in air.

with the decrease of the air gap below 6 mm, which is the maximum travel distance of the tritium beta particles. At an air gap of 0.1 mm, the saturation current density dropped to about half of the value, thus, the CPD cell collected about 40% of current here. The maximum power point yields 1.3 nW (0.4 nW/cm²), which corresponds to a fill factor of 0.3. The current may be increased by increasing the gas pressure or/and by using a gas with a lower ionization energy, such as argon.

Both the open circuit voltage V_{oc} and the short circuit current I_{sc} decrease with electrode separation d (see Fig. 3). The short circuit current shows a more pronounced drop with distance than the open circuit voltage. This is reasonable considering that the maximum range of tritium beta particles in air is about 6 mm and that their kinetic energy decays exponentially with the distance traversed. Moreover, electron-ion recombination increases as the electrode separation increases.

In an attempt to further enhance the energy conversion, solid state CPD cells were fabricated and tested.

Intrinsic hydrogenated amorphous silicon (a-Si:H) was used as the dielectric medium, sandwiched between a platinum (Pt) anode and ScT cathode. Titanium and platinum were sequentially evaporated on a silicon substrate using an e-beam evaporator. The Ti and Pt films were approximately 20 and 200 nm thick, respectively. Subsequently, a-Si:H film of 400 nm thickness was deposited on the Pt film using the dc saddle field plasma-enhanced chemical-vapor deposition technique.^{2,13} The deposition was carried out by decomposing pure silane gas in a glow discharge. The flow rate of silane was 30 SCCM (SCCM denotes cubic centimeter per minute at STP). The deposition chamber pressure was 160 mTorr. The anode potential was 600 V. The anode current was 18 mA. The ion current at the substrate was 0.6 mA. The substrate temperature was 250 °C. The deposition time was 1 h and 20 min. An array of scandium dots was thermally evaporated onto the a-Si:H. The dots were of 3 mm diameter and 300 nm thickness, each.

A schematic of the CPD cell is illustrated in the inset of Fig. 4. Prior to the tritiation step, each cell was tested for its *IV* characteristics under darkness and room light illumination. The cells showed good rectification and photovoltaic characteristics. The tested CPD cells were then exposed in tritium gas for 10 h at a temperature of 250 °C and a pressure of 10 bars. The resulting scandium tritide film showed a surface activity of 15 mCi/cm², as discussed earlier. In



FIG. 4. (Color online) The IV characteristics of a solid CPD cell in darkness before tritiation and betavoltaic performance after tritiation. The solid dielectric is a-Si:H.

a-Si:H, the mean ionization energy is reported to be approximately 4.5 eV per electron-hole pair.¹⁰ The theoretical upper limit for the short circuit current density can be estimated using the following relation:¹⁰

$$J_{\rm sc} = \lambda n_T E_{\beta} / \varepsilon, \tag{1}$$

where λ is the decay constant of tritium $(1.78 \times 10^{-9} \text{ s})$, n_T is the surface activity of the scandium tritide film (15 mCi/cm², which is equivalent to $3.1 \times 10^{17} \text{ atoms/cm}^2$), E_{β} is the average kinetic energy of the tritium betas (5.7 keV), and ε represents the ionization energy (4.5 eV). Equation (1) yields a J_{sc} of 110 nA/cm².

After the tritiation, dark *IV* measurements were carried out. A typical result is shown in Fig. 4. Powered by the ScT film, an open circuit voltage of 160 mV and a short circuit current of 370 pA were measured. The area of the ScT disk is about 7 mm². The maximum power point was measured to be 18 pW (0.26 nW cm²). The short circuit current density was measured to be 5.3 nA/cm², which was 4.8% of the theoretical limit calculated by Eq. (1). Owing to a smaller ionization energy in *a*-Si:H, the short circuit current density was doubled compared to that of the cell with the gaseous dielectric, though the overall output power density was not improved owing to the lower open circuit voltage.

Both CPD cells with the gaseous and semiconductor dielectrics produced electrical power at the nanowatt level. The energy conversion efficiency is found to be about 0.1%. This is comparable to other reported betavoltaic devices powered by tritium^{2,4} and less than that of the recently reported SiC pn-junction cell using high energy betas from ⁶³Ni and ³³P.^{5,6} In the gaseous CPD cells, the low-power density and conversion efficiency is attributed to the relatively high ionization energy, while in the *a*-Si:H CPD cells, the lower performance is attributed to the low built-in field and associated higher recombination rate.

In summary, tritium based nuclear batteries using ScT_r films have been demonstrated. The batteries utilize air and a-Si: H film as dielectric media, displaying short circuit current densities of 40% and 4.8% of the theoretical limits, respectively. The batteries provide power with a projected lifetime comparable to the half-life of tritium and can be operated at temperatures approaching 250 °C. The appropriateness of ScT_x is also evident from its low tritium vapor pressure at elevated temperatures.¹⁴ The output power density can be further enhanced by increasing the surface activity of the ScT_x film using a textured surface as well as through improved passivation of the CPD cell. It is suggested that using a stacked series-connected cell geometry, a tritium battery based on the CPD principle could be realized with a continuous output power of 1 μ W at a cell volume of $\sim 1 \text{ cm}^3$.

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- ¹K. E. Bower, Y. A. Barbanel, Y. G. Shreter, and G. W. Bohnert, *Polymer*, *Phosphors, and Voltaics for Radioisotope Microbatteries* (CRC, Boca Raton, FL, 2002).
- ²T. Kosteski, N. P. Kherani, P. Stradins, F. Gaspari, W. T. Shmayda, L. S. Sidhu, and S. Zukotynski, IEE Proc.: Circuits Devices Syst. **150**, 274 (2003).
- ³A. Lal and J. Blanchar, IEEE Spectrum **41**, 36 (2004).
- ⁴W. Sun, N. P. Kherani, K. D. Hirschman, L. L. Gadeken, and P. M. Fauchet, Adv. Mater. (Weinheim, Ger.) **17**, 1230 (2005).
- ⁵M. V. S. Chandrashekhar, C. I. Thomas, H. Li, M. G. Spencer, and A. Lal, Appl. Phys. Lett. **88**, 033506 (2006).
- ⁶C. J. Eiting, V. Krishnamoorthy, S. Rodgers, T. George, J. D. Robertson, and J. Brockman, Appl. Phys. Lett. **88**, 064101 (2006).
- ⁷B. Liu, K. P. Chen, N. P. Kherani, T. Kosteski, S. Costea, S. Zukotynski, and A. B. Antoniazzi, Appl. Phys. Lett. **89**, 044104 (2006).
- ⁸B. Liu, K. P. Chen, N. P. Kherani, S. Zukotynski, and A. B. Antoniazzi, Appl. Phys. Lett. 88, 134101 (2006).
- ⁹J. R. Srour, C. J. Marshall, and P. W. Marshall, IEEE Trans. Nucl. Sci. **50**, 653 (2003).
- ¹⁰T. Kosteski, Ph.D thesis, University of Toronto, 2001.
- ¹¹N. P. Kherani, B. Liu, K. Virk, T. Kosteski, F. Gaspari, W. T. Shmayda, S. Zukotynski, and K. P. Chen, J. Appl. Phys. **103**, 024906 (2008).
- ¹²N. P. Kherani and W. T. Shmayda, Fusion Technol. 28, 893 (1995).
- ¹³N. P. Kherani, T. Kosteski, S. Zukotynski, and W. T. Shmayda, Fusion Technol. 28, 1609 (1995).
- ¹⁴A. W. P. Poon, R. J. Komar, C. E. Waltham, M. C. Browne, R. G. H. Robertson, N. P. Kherani, and H. B. Mak, Nucl. Instrum. Methods Phys. Res. A **452**, 115 (2000).

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